

# High electric field performance of $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}$ and $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}/\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$ quantum well micro-Hall devices

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## Abstract

Quantum well micro-Hall devices based on uniformly Si-doped  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}$  and Si- $\delta$ -doped  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}/\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$  heterostructures are investigated as function of electric field and compared in terms of sensitivity and noise properties. The data show that at high electric fields, doped-channel quantum well devices are advantageous over high-mobility structures and that the use of pseudomorphic InGaAs results in better performance than does GaAs. A maximal signal-to-noise sensitivity (SNS) of  $138 \text{ dB T}^{-1}$  is achieved in a  $10 \mu\text{m} \times 10 \mu\text{m}$  device at 300 K, at frequency of 100 kHz and bandwidth of 1 Hz. This performance corresponds to a lowest detection limit of  $127 \text{ nT Hz}^{-1/2}$ , with no degradation for electric fields up to  $2.4 \text{ kV cm}^{-1}$ ; these values represent the best reported at such high electric fields. Furthermore, our results suggest that a signal-to-noise sensitivity of  $160 \text{ dB T}^{-1}$  and a lowest detection limit of 10 nT is achievable in doped-channel structures.

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## 1. Introduction

The precise study of the material's surface magnetic structure by means of scanning Hall probe microscopy (SHPM) [1], scanning superconducting quantum interference device susceptometers (SQUID) [2] and magnetic force microscopies (MFM) [3] requires magnetic sensors of high sensitivity and signal linearity over a wide range of magnetic fields and temperatures as well as small active areas. To meet the demands of these experimental techniques, advanced III–V micro-Hall sensors with a high sensitivity, low noise, high thermal stability, and low temperature dependence are required. Modulation-doped heterostructures with high electron mobility and low electron concentration can be used to achieve very high magnetic sensitivities, but suffer from thermal drift and trapping effects which cause noise.

Furthermore, the performance of these structures is very temperature dependent. In Hall devices, small enough to probe the microstructure of magnetic materials, the electric field will normally be quite large. For sensor applications with high spatial resolution, requirements include maximum signal-to-noise sensitivity (SNS), good temperature stability, and little trapping even under high electric fields.

In the last years, Hall sensors using modulation-doped AlGaAs/GaAs [4], AlAs/GaAs [5], AlGaAs/InGaAs/GaAs [6], InAlAs/InGaAs/InP [7] and Si- $\delta$ -doped AlGaAs/InGaAs/GaAs [8] or InAs/AlGaSb [9] QWs have been proposed. The high electron mobility of the 2DEG is advantageous primarily within the low-electric field region [4]. At higher electric fields ( $E > 1 \text{ kV cm}^{-1}$ ), the electron mobility in modulation-doped structures decreases significantly [5]. As a result, the supply-voltage-related sensitivity noticeably degrades. Additionally, the sheet carrier concentration in modulation-doped AlGaAs/GaAs heterostructures often increases with rising electric field, leading to a decrease in the supply-current-related sensitivity. More critical, because the mobility is

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highly temperature dependent in high-mobility modulation-doped structures, measurements made as a function of temperature at lower temperatures require a constant recalibration of the Hall sensor.

In this work, we describe and compare QW micro-Hall devices fabricated with uniformly Si-doped GaAs QWs in AlGaAs/GaAs heterostructures and Si- $\delta$ -doped  $\text{In}_y\text{Ga}_{1-y}\text{As}$ :Si strained QWs in AlGaAs/GaAs/InGaAs heterostructures. We show that under larger electric fields, the signal-to-noise sensitivity is determined to a great extent by the maximum carrier velocity in the structure, and only to a lesser degree by carrier concentration and mobility. Thus, performance is optimized with materials with high electron saturation velocity, high electron concentration, and excellent materials quality leading to low  $1/f$  noise. Furthermore, very good signal-to-noise performance is achievable at high electric fields due to the higher velocity at these fields. Our data further show that the  $1/f$  noise also increases at higher electric fields, requiring that higher frequencies be used at higher electric fields. In our work, we achieve a maximal signal-to-noise sensitivity of  $138 \text{ dB T}^{-1}$  in a  $10 \mu\text{m} \times 10 \mu\text{m}$  device at 300 K with bandwidth of 1 Hz, corresponding to a lowest detection limit of 127 nT at a frequency of 100 kHz at an electric field of  $2.4 \text{ kV cm}^{-1}$ . We project that at higher measurement frequencies and higher fields, values of  $160 \text{ dB T}^{-1} \text{ Hz}^{1/2}$  and  $10 \text{ nT Hz}^{1/2}$ . Using the figure of merit, signal-to-noise sensitivity scaled by the width of the active region, we expect a value of  $200 \text{ dB T}^{-1} \text{ Hz}^{1/2} \text{ mm}^{-1}$ .

## 2. Device fabrication

Using a Riber 32-P gas-source molecular-beam epitaxy system,  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}$  and  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}/\text{GaAs}/\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$  QWs were grown on semi-insulating (001)-oriented GaAs wafers. For the group III elements Al, Ga, and In, solid sources were used; the group V element As was obtained from a  $\text{AsH}_3$  hydride gas source, cracked at  $830^\circ\text{C}$ . The surface quality was monitored by RHEED pattern both during the oxide desorption and subsequent growth. The substrate temperature was controlled using a thermocouple and an infrared pyrometer.

The epilayer sequences for the two samples primarily discussed in this paper are given in Table 1. The 1000 Å GaAs buffer layer initially grown on the substrate ensures high material and interface quality of all subsequent layers. The GaAs was grown at a growth rate of  $2.0 \text{ \AA s}^{-1}$  at a growth temperature of  $575^\circ\text{C}$ . The  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$  barriers were grown at  $630^\circ\text{C}$ . In structure A, a uniformly-doped GaAs:Si layer was inserted between adjacent  $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$  barriers, serving as a conductive channel for the 2DEG. The structure was capped with heavily-doped GaAs:Si, which contributes to better Ohmic contacts.

Structure B has essentially the same layer sequence as in structure A, but the addition of a 30 Å Si- $\delta$ -doped

Table 1  
Structural and electronic parameters of the two samples studied

|                                                                              |                                                            | Sample |      |
|------------------------------------------------------------------------------|------------------------------------------------------------|--------|------|
|                                                                              |                                                            | A      | B    |
| GaAs:Si (cap layer)                                                          | $d$ (nm)                                                   | 7      | 7    |
|                                                                              | $N_d$ ( $10^{18} \text{ cm}^{-3}$ )                        | 2      | 2    |
| $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$ (barrier)                          | $d$ (nm)                                                   | 35     | 35   |
| GaAs:Si (quantum well)                                                       | $d$ (nm)                                                   | 17.5   | 16   |
|                                                                              | $N_d$ ( $10^{18} \text{ cm}^{-3}$ )                        | 2      | 2    |
| Si- $\delta$ -doped $\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$ (quantum well) | $d$ (nm)                                                   |        | 3    |
|                                                                              | $n_{2d}$ ( $10^{11} \text{ cm}^{-2}$ )                     |        | 2.1  |
| GaAs:Si (quantum well)                                                       | $d$ (nm)                                                   | 17.5   | 16   |
|                                                                              | $N_d$ ( $10^{18} \text{ cm}^{-3}$ )                        | 2      | 2    |
| $\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$ (barrier)                          | $d$ (nm)                                                   | 35     | 35   |
| GaAs:Be                                                                      | $d$ (nm)                                                   | 10     | 10   |
|                                                                              | $N_a$ ( $10^{18} \text{ cm}^{-3}$ )                        | 1.5    | 1.5  |
| GaAs (buffer layer)                                                          | $d$ (nm)                                                   | 100    | 100  |
| GaAs                                                                         | S.I. substrate                                             |        |      |
| Hall data (at 300 K)                                                         | $n_s$ ( $10^{12} \text{ cm}^{-2}$ )                        | 1.28   | 1.5  |
|                                                                              | $\mu_{dc}$ ( $\text{cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ) | 1651   | 2046 |

$\text{In}_y\text{Ga}_{1-y}\text{As}$ :Si strained QW inserted into the GaAs:Si channel layer. The  $\delta$ -doping position is centered within this QW. The  $\text{In}_y\text{Ga}_{1-y}\text{As}$  layer was grown at a substrate temperature of  $565^\circ\text{C}$ . The resulting layer thickness was kept well below the critical layer thickness (CLT) [10–15] in order to avoid misfit dislocations. Structural quality was monitored during growth using RHEED techniques.

Using a standard optical photolithography, four-terminal Greek-cross micro-Hall devices with active area of  $10 \mu\text{m}$  square size were fabricated. Alloying an evaporated Au:Ge:Ni metal film, Ohmic contacts were formed, with measured contact resistances of 7 and  $14 \Omega$  for Samples A and B, respectively. Chemical wet-etching was used for mesa device isolation. Varying the temperature for the thermal annealing procedure in the range of  $320\text{--}450^\circ\text{C}$ , we found the optimal conditions for lowest specific resistance (20 s at  $420^\circ\text{C}$ ). Finally, individual chips were cut from the die, mounted and bonded on a non-magnetic chip carrier.

Both Hall mobilities and carrier concentrations in the conductive channel were also measured using the van der Pauw method in large-sized samples.

## 3. Magnetic sensitivity

The Hall voltage is given by

$$V_H = G \frac{IB}{en_s} \quad (1)$$

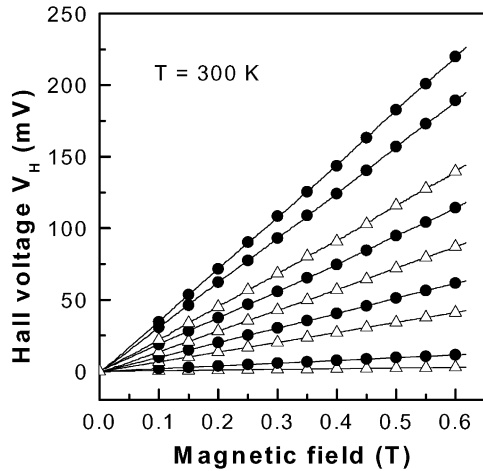


Fig. 1. Hall voltages as functions of magnetic induction in current drive mode at  $T = 300$  K for different values of the bias current—( $\Delta$ ): Sample A (10, 120, 250, and  $400 \mu\text{A}$ ); ( $\bullet$ ): Sample B (50, 200, 400, 800, and  $1000 \mu\text{A}$ ).

where  $I$  is the current,  $B$  the perpendicular magnetic field,  $e$  the electronic charge,  $n_s$  the sheet electron concentration, and  $G$  is the geometric correction factor. For our Greek-cross structures, where length of the active square is about one-third of the distance between the contacts,  $G \approx 1$  [16]. Thus,

$$V_H \approx \mu EWB = vWB \quad (2)$$

where  $\mu$  and  $v$  are the electron mobility and velocity,  $E$  the electric field in the active region of the Hall device, and  $W$  the width of the active (square) region. Fig. 1 shows the Hall voltage for our micro-Hall elements A and B at  $T = 300$  K as function of  $B$  for various currents (and, therefore, various electric fields). Both devices exhibit an excellent linearity over an interesting range of magnetic field. The residual non-linearity of the micro-Hall device B is seen in Fig. 2 for both low and high bias currents. The estimated non-linearity does

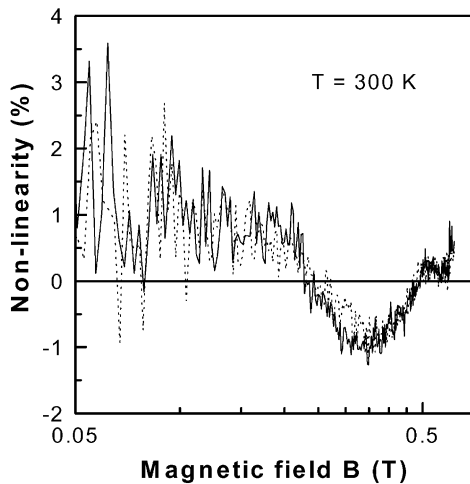


Fig. 2. Non-linearity of the Hall voltage  $[(V_H(I, B) - V_{H, \text{fit}})/V_{H, \text{fit}}]$  vs. magnetic field at  $T = 300$  K for Sample B: dotted line,  $I = 10 \mu\text{A}$ ; full line,  $I = 900 \mu\text{A}$ .

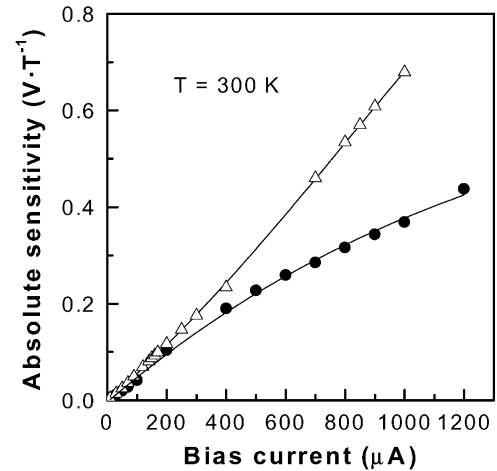


Fig. 3. Absolute sensitivity as function of bias current for Samples A and B at  $T = 300$  K—( $\Delta$ ): Sample A, ( $\bullet$ ): Sample B.

not exceed 4% and is mainly caused by measurement errors. Similar results were obtained for Sample A.

The slope of the Hall voltage plotted against  $B$  is the absolute sensitivity,

$$S_A = \frac{dV_H}{dB} \approx \mu EW = vW \quad (3)$$

Fig. 3 shows the absolute sensitivity as function of bias current for Hall devices fabricated from both structures at room temperature. For Sample A, the largest measured value of  $S_A = 0.679 \text{ V T}^{-1}$  for a bias current of  $I = 1000 \mu\text{A}$ , corresponding to an electric field of about  $5 \text{ kV cm}^{-1}$ . For Sample B, the value of  $S_A = 0.438 \text{ V T}^{-1}$  was measured for a current of  $I = 1200 \mu\text{A}$  corresponding to an electric field of  $3.1 \text{ kV cm}^{-1}$ .

The ratio of the absolute sensitivity to current is the current-related sensitivity,

$$S_I = \frac{1}{I} \frac{dV_H}{dB} \approx \frac{1}{n_s e} \quad (4)$$

Because of the dependence of  $S_I$  on  $n_s$ , the current-related sensitivity of Sample A is higher than that of Sample B. As will be discussed later, a low  $n_s$  has a detrimental effect on the noise. The non-constant slopes of  $S_A$  seen in Fig. 3 reflects the variations in  $S_I$  due to changes in  $n_s$ . For our micro-Hall device A, we derived best values of  $S_I = 590 \text{ V A}^{-1} \text{ T}^{-1}$  at room temperature and  $S_I = 565 \text{ V A}^{-1} \text{ T}^{-1}$  at  $T = 77$  K, respectively. For Sample B, the maximum of the absolute sensitivity is found to be  $S_A = 0.454 \text{ V T}^{-1}$  at  $T = 77$  K and  $S_A = 0.438 \text{ V T}^{-1}$  at  $T = 300$  K.

Hall sensors based on modulation-doped AlGaAs/GaAs and AlAs/GaAs heterostructures can have very high values of  $S_I$  when  $n_s$  is made small; values as high as  $1200 \text{ V A}^{-1} \text{ T}^{-1}$  have been reported [4,5]. Furthermore, the voltage-related sensitivity

$$S_V = \frac{1}{V} \frac{dV_H}{dB} \approx \mu \quad (5)$$

is also extremely high due the high values of the electron mobility,  $\mu$ . On the other hand, DX centers are responsible for a number of disadvantages in such structures, including extremely large changes in carrier concentration at higher electric fields [5].

Structures A and B show almost no dependence of  $n_{2d}$  on temperature; the mobilities increase somewhat with increasing temperature, reflecting the positive temperature coefficient for mobility dominated by ionized impurity scattering. The 77 K mobility also increases somewhat with increasing electric field, since the electric field heats the electrons and, again, higher electron temperature leads to reduced ionized impurity scattering. The effect of the higher electric field on the carrier concentration is that it becomes somewhat smaller. This effect results in an increase in  $S_I$ . For Sample A, the low-electric field  $S_I$  is  $S_I = 470 \text{ V A}^{-1} \text{ T}^{-1}$ ; this sensitivity increases between fields of 0 and  $600 \text{ V cm}^{-1}$ , reaching  $S_I = 590 \text{ V A}^{-1} \text{ T}^{-1}$  at  $600 \text{ V cm}^{-1}$  and above. At 77 K, the value of  $S_I$  varies between  $S_I = 520 \text{ V A}^{-1} \text{ T}^{-1}$  at low fields and  $S_I = 565 \text{ V A}^{-1} \text{ T}^{-1}$  at fields above about  $1.6 \text{ kV cm}^{-1}$ . For Sample B at 300 K, the low-electric field is  $S_I = 390 \text{ V A}^{-1} \text{ T}^{-1}$ , increasing to  $S_I = 520 \text{ V A}^{-1} \text{ T}^{-1}$  at  $350 \text{ V cm}^{-1}$ , and then falling again to  $S_I = 390 \text{ V A}^{-1} \text{ T}^{-1}$  at  $1.5 \text{ kV cm}^{-1}$ . At 77 K, the value of  $S_I$  varies between  $S_I = 415 \text{ V A}^{-1} \text{ T}^{-1}$  at low fields and  $S_I = 450 \text{ V A}^{-1} \text{ T}^{-1}$  at fields above about  $1.8 \text{ kV cm}^{-1}$ . The mechanism of free electron depletion within the QW is assumed to be a thermal or field induced trap of carriers by DX centers. The lifetime of captured carriers is normally high for these centers and depends substantially on alloy composition, electron concentration and temperature [17,18]. An estimation of this lifetime can be also derived from noise measurements as shown below.

#### 4. Noise spectra

Measurements of voltage fluctuations across the Hall electrodes have been carried out at zero magnetic field as a function of frequency. The value of noise is expected to be the same in the low range of magnetic fields where the condition  $\mu^2 B^2 \ll 1$  is satisfied [7]. The measurements were made using different directions of current flow and various contacts. The typical noise spectra for our micro-Hall elements A and B at low-electric field are shown in Fig. 4.

At the lowest measured frequencies, all spectra have a tendency to show a plateau. We have measured noise spectra in a wider range of frequencies down to 0.027 Hz to confirm this tendency for our micro-Hall device B. We attribute this low-frequency noise to generation and recombination. Generation-recombination noise can be expressed as:

$$S_{V,g-r}(f) \propto \frac{1}{1 + (2\pi f\tau)^2} \quad (6)$$

where  $\tau$  is the characteristic time constant. Evaluating the low-frequency data, we obtain 1.2 and 140 ms for Samples

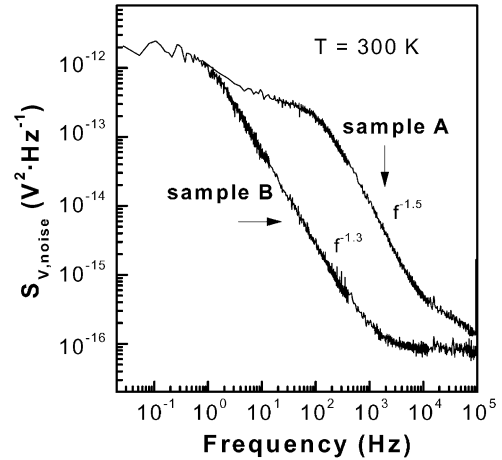


Fig. 4. Noise voltage spectra across the Hall electrodes for Samples A and B at a bias current of  $10 \mu\text{A}$  at  $T = 300 \text{ K}$ .

A and B, respectively. Such anomalous high magnitudes of the time constant are typical for DX centers [18].

Between 1 Hz and about 10 kHz, the noise spectrum is dominated by flicker noise with its characteristic  $1/f$  behavior. Empirically, the actual form of the noise spectral density is  $1/f^n$ , with  $n = 1.5$  and  $1.3$  for Samples A and B, respectively. These curves can be fit as the sum of generation-recombination noise with its  $1/f^2$  frequency dependence and flicker noise with a purely  $1/f$  behavior. (Similar behavior was observed by Sugiyama et al. [5] who concluded, however, that the slope in the noise spectrum of 2DEG Hall elements was related to the processing of the Ohmic contacts.) The voltage noise spectral density for  $1/f$  component of the mid-frequency noise,  $S_{V,\alpha}$  across the contacts measuring the Hall voltage is given by

$$S_{V,\alpha} = \alpha E^2 (n_{2d} f)^{-1} \quad (7)$$

where  $\alpha$  is the Hooge parameter for the material and  $E$  the applied electric field in the active area of the Hall device. Fitting the measured noise spectral density for the two devices as being the sum of generation-recombination type noise and  $1/f$  noise, the Hooge parameters for the two structures were extracted. We find that for Sample A at 300 K,  $\alpha = 1.5 \times 10^{-4}$  and for Sample B, also at 300 K,  $\alpha = 1.1 \times 10^{-4}$ . These values are typical for these materials doped as these are.

At frequencies high enough that the  $1/f$  noise is small, the noise spectrum is dominated by thermal noise, which is independent of frequency. In Sample B, thermal noise dominates for frequencies above about 5 kHz; in Sample A, because of its higher  $1/f$  noise, the thermal noise dominates for frequencies above about 50 kHz. The spectral noise density for thermal noise is given by

$$S_{V,\text{thermal}} = 4k_B T R \quad (8)$$

where  $k_B$ ,  $T$ , and  $R$  are the Boltzmann constant, the absolute temperature and the resistance of the semiconductor between the two Hall voltage contacts, respectively. For

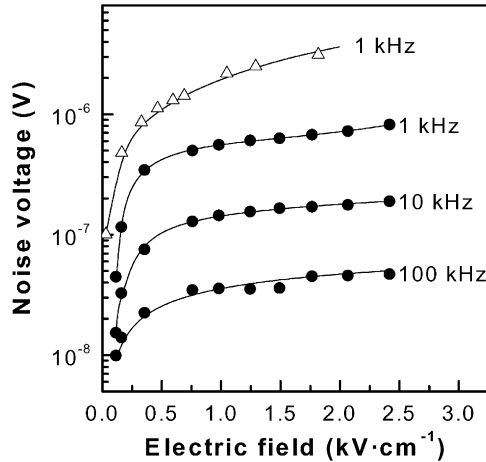


Fig. 5. Noise voltage as a function of applied electric field for micro-Hall devices A ( $\triangle$ ) and B ( $\bullet$ ),  $\Delta f = 1$  Hz,  $T = 300$  K.

Sample B, one calculates a value of  $S_{V,\text{thermal}} = 8.4 \times 10^{-17} \text{ V}^2 \text{ Hz}^{-1}$ , which is in excellent agreement with the measured value seen in Fig. 4. The value for Sample A is about 50% higher, but not clearly resolved still at 100 kHz.

The noise voltage in a band of frequencies,  $\Delta f$ , is given by

$$\bar{V}_{\text{noise}} = \sqrt{S_{V,\text{noise}}(f)\Delta f} \quad (9)$$

where  $S_{V,\text{noise}}$  is the voltage noise spectral density. Increasing the bias current and consequently the strength of electric field leads to an increase of noise voltage spectral density and, therefore, to the noise voltage (see Fig. 5). As long as the noise is dominated by flicker noise, the noise voltage scales approximately linearly with electric field. In fact, the noise voltage of Sample A measured at 1 kHz is proportional to  $E$ ; the dependencies of Sample B are superlinear at small fields, but sub-linear at large fields. The thermal noise, on the other hand, is expected to increase only very little, due to the increase in the resistance as the velocity saturates.

The SNS sensitivity is defined as the absolute sensitivity divided by the noise voltage for a given bandwidth,

$$\text{SNS} \equiv \frac{S_A}{\bar{V}_{\text{noise}}} \quad (10)$$

Its dependence on electric field is shown in Fig. 6 for several frequencies and using a measurement bandwidth of  $\Delta f = 1$  Hz. Because at most electric fields, the noise is still dominated by  $1/f$  noise, the SNS is approximately constant, since both  $S_A$  and  $\bar{V}_{1/f}$  increase approximately linearly with increasing electric field.

As seen from Fig. 6, the SNS of our micro-Hall elements does not degrade even at enough high electric fields ( $E > 1.5 \text{ kV cm}^{-1}$ ). The maximum SNS of  $80 \text{ dB T}^{-1}$  at 10 Hz and  $98 \text{ dB T}^{-1}$  at 1 kHz for the micro-Hall device A are achieved. For the micro-Hall device B, the maximum

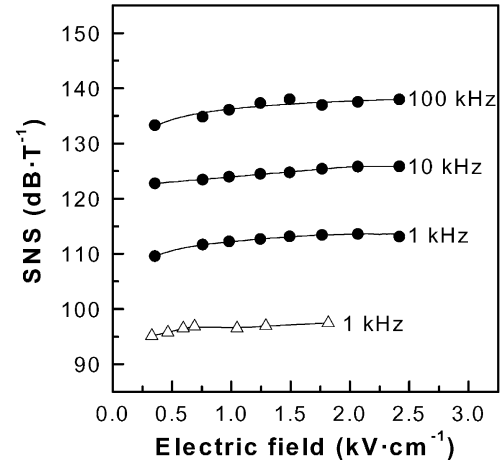


Fig. 6. Signal-to-noise sensitivity as function of applied electric field at room temperature for micro-Hall devices A ( $\triangle$ ) and B ( $\bullet$ ).

magnitudes of SNS are  $88 \text{ dB T}^{-1}$  at 10 Hz,  $114 \text{ dB T}^{-1}$  at 1 kHz, and  $138 \text{ dB T}^{-1}$  at 100 kHz.

We can estimate the expected values of SNS for the two cases of  $1/f$  noise and thermal noise for a cross-shaped Hall device with width  $W$  measured in the frequency band width  $\Delta f$  as

$$\text{SNS} \approx \mu W \sqrt{\frac{fn_s}{\alpha}} \sqrt{\frac{1}{\Delta f}} \quad (11)$$

and

$$\text{SNS} \approx v W \sqrt{\frac{en_s \mu}{4k_B T}} \sqrt{\frac{1}{\Delta f}} \quad (12)$$

respectively. We see that at middle frequencies, where  $1/f$  noise dominates, the SNS can be approximately constant in electric field, as long as the electron mobility remains constant. At the highest frequencies, where thermal noise dominates, the SNS can continue to increase until velocity saturation is reached. In the case of our doped structures, the electron mobility is dominated at low temperatures by ionized impurity scattering and at room temperature by both ionized impurity scattering and phonon scattering. Thus, in contrast to high-mobility structures, the mobility is relatively constant, increasing somewhat with increasing electron temperature. This effect has been seen in detailed investigations of hot electron velocity at high electric field strengths in doped quantum wells [19,20].

At the same time, the investigated structures show a decrease of the electron concentration due to trapping effect on DX centers. Consequently, the decrease of SNS is obvious (see Eqs. (11) and (12)) depending on changes of the electron concentration. Even for such an electron concentration behavior we have shown that SNS does not degrade at electric field up to  $2.5 \text{ kV cm}^{-1}$ . Furthermore, this problem can be decided in different ways: (a) by means of using  $\delta$ -doped  $\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$  QWs spaced from the

$\text{Al}_{0.3}\text{Ga}_{0.7}\text{As}$  barrier by undoped GaAs or (b) due to the smaller Al content of  $\delta$ -doped  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$  and  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}/\text{In}_y\text{Ga}_{1-y}\text{As}$  heterostructures. That will be the subject of our following studies.

We estimate the detection limit at different frequencies as

$$B_{\text{DL}} = \left[ \frac{B_{v,\text{noise}}(f)\Delta f}{S_A^2} \right]^{1/2} = \frac{1}{\text{SNS}} \quad (13)$$

The micro-Hall sensors based on  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$  heterostructures show a lowest detection limit of 103 and 14  $\mu\text{T}$  at 10 and 1 kHz, respectively for a bandwidth of 1 Hz. The micro-Hall sensors with the InGaAs active channel have a lowest detection limit of 41 and 2  $\mu\text{T}$  at 10 and 1 kHz, respectively.

A detailed analysis of the detection limit dependence on the applied electric field as a function of frequency is presented in Fig. 7 for our micro-Hall device B. It is clearly seen that the lowest detection limit can be achieved at higher electric fields and at higher operation frequencies. The lowest measured detection limit for a 1 Hz bandwidth is 127 nT at an operation frequency of 100 kHz and an applied electric field of 2.4  $\text{kV cm}^{-1}$ . Since the absolute sensitivity is proportional to the electron velocity, the highest absolute sensitivity will be reached at high electric fields, when the material is in velocity saturation. For doped pseudomorphic InGaAs, this value can reach  $2 \times 10^7 \text{ cm s}^{-1}$ , allowing an absolute sensitivity of  $S_A = 2 \text{ V T}^{-1}$ . On the other hand, at high electric fields, the  $1/f$  noise also increases, which is why the signal-to-noise sensitivity seen in Fig. 6 does not increase much with increasing electric field. The data imply, however, that at sufficiently high frequencies, also this increased  $1/f$  noise falls off and only the thermal noise remains at a value of not higher than  $2 \times 10^{-8} \text{ V}$  for a 1 Hz measurement bandwidth. Thus, for high frequencies, we anticipate a signal-to-noise sensitivity of 160  $\text{dB T}^{-1}$  (or expressed with arbitrary bandwidth, 160  $\text{dB T}^{-1} \text{ Hz}^{1/2}$ ) at 300 K for  $10 \mu\text{m} \times 10 \mu\text{m}$  Hall devices. For comparison

purposes, a useful figure of merit is the signal-to-noise sensitivity scaled by the width of the active region. One sees from both Eqs. (11) and (12) that the SNS scales not only with  $\Delta f^{-1/2}$ , but also with  $W$ . We project that the ultimate scaled SNS for the doped-channel InGaAs is at least 200  $\text{dB T}^{-1} \text{ Hz}^{1/2} \text{ mm}^{-1}$ .

## 5. Conclusion

Highly sensitive GaAs and InGaAs 2DEG micro-Hall devices have been fabricated from  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$  and  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}/\text{In}_y\text{Ga}_{1-y}\text{As}$  heterostructures. These devices exhibit an excellent linearity at electric fields higher than 1.8  $\text{kV cm}^{-1}$ . Micro-Hall elements based on  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$  and  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}/\text{In}_y\text{Ga}_{1-y}\text{As}$  heterostructures show a magnetic sensitivity of 590 and 517  $\text{V A}^{-1} \text{ T}^{-1}$  and a signal-to-noise sensitivity of 98 and 114  $\text{dB T}^{-1}$ , respectively, not degrading even at high electric field strength. The micro-Hall devices based on  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}/\text{In}_y\text{Ga}_{1-y}\text{As}$  QWs show a lower value of voltage noise spectral density in comparison with micro-Hall devices based on a  $\text{Al}_x\text{Ga}_{1-x}\text{As}/\text{GaAs}$  heterostructure. In this case, a lowest detection limit of 127 nT at a frequency of 100 kHz and electric fields of about 2.4  $\text{kV cm}^{-1}$  was determined, never achieved before at electric fields  $E > 1.5 \text{ kV cm}^{-1}$ .

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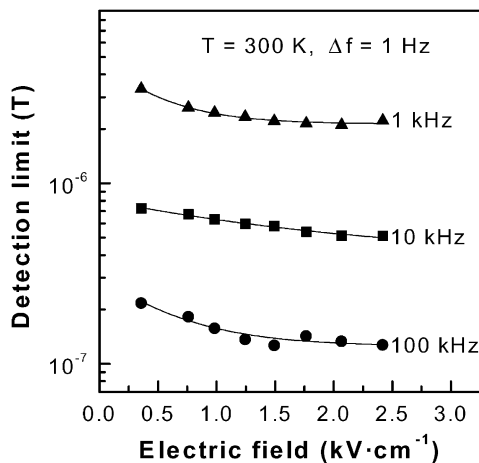


Fig. 7. Detection limit for Sample B as function of applied electric field measured at several frequencies.

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